# NMR signatures of polymer chain dynamics in bulk and under nanoscopic confinements

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# Outline

- Short introduction to NMR methods
- Polymer dynamics theories
- Bulk polymer melts
- Confined polymer melts
- Simulations



assume: magnetization tilted against  $\vec{B}_{n}$ 

 $\rightarrow$  precesses with a certain Larmor frequency

and a certain precession phase



How can we generate such "spin coherences"?



Free induction signal:

### magnetization precesses

## => flux in an RF pick-up coil oscillates

## => signal voltage is induced





## NMR measurands of interest:

a) the spin-lattice relaxation time  $T_I$ 

... is the time constant with which thermal equilibrium of the longitudinal magnetization is approached ... is due to energy transfer

between spins and mechanical degrees of freedom

... is caused by induced spin transitions "up"  $\rightarrow$  "down" and "down"  $\rightarrow$  "up"

... reflects motions of chain segments at the resonance frequency

b) the mean square displacement  $\langle R^2 \rangle$ 

... is the second moment of the probability density function ... is due to self-diffusion of chain segments

## NMR

	10 <sup>1</sup>	10 <sup>2</sup>	10 <sup>3</sup>	10 <sup>4</sup>	10 <sup>5</sup>	10 <sup>6</sup>	10 <sup>7</sup>	10 <sup>8</sup>	$10^9 \frac{a}{rad}$	o Hz
	field cyc					ng relax	ometry			
			ro	rot. frame relax.				conv. relaxometry		
	diffusometry, transverse relaxation, residual spin couplings									
<u>τ</u>	10 <sup>-1</sup>	10 <sup>-2</sup>	10 <sup>-3</sup>	10-4	10 <sup>-5</sup>	10-6	10-7	10 <sup>-8</sup>	10 <sup>-9</sup>	

S

## translational diffusion

spin-lattice relaxation

rotational diffusion of nuclear magnetic dipole pairs or electric quadrupoles in anisotropic heterogeneous media

# Spin-lattice relaxometry





## Field-gradient NMR diffusometry





- reversible refocusing of the transverse magnetization in an inhomogeneous magnetic field
- refocusing incomplete in the presence of translational diffusion
- echo attenuation indicative for translational diffusion

**Diffusion measurements in the fringe field** of an ordinary superconducting magnet



three model theories for polymer chain modes:

Rouse model

(chain in a viscous medium; no hydrodynamic backflow; no "entanglements" i.e.  $M < M_c$ )

- Renormalized Rouse formalism (chains subject to "entanglements" i.e. M>M<sub>c</sub>)
- Tube/reptation concept (chains confined in nanoscopic tubes)

# Rouse model

### Rouse model:

Bead-and-spring chain in a viscous medium without backflow

$$(M < M_c)$$

equation of motion for the *n*-th bead:

$$K(2\vec{r_n} - \vec{r_{n+1}} - \vec{r_{n-1}}) - \zeta \frac{\partial \vec{r_n}}{\partial t} + \vec{F_n} \approx \frac{K}{\partial n^2} \frac{\partial^2 \vec{r_n}}{\partial t} - \zeta \frac{\partial \vec{r_n}}{\partial t} + \vec{F_n} = 0$$

(entropic spring const.  $K = \frac{3k_BT}{b^2}$ ; friction coeff.  $\zeta = 6\pi\eta a_h$ ; random force  $\vec{F_n}$ ) Solution: Superposition of discrete Rouse relaxation modes with time constants

$$T_p = \frac{\zeta b^2 N^2}{3\pi^2 k_B T p^2}$$
, where  $p = 1 \cdots N$ 

P. E. Rouse, J. Chem. Phys. 21 (1953) 1272

## NMR measurand laws predicted by the Rouse model:

(polymer chains in a viscous medium without "entanglements" and hydrodynamic backflow)



( $\tau_s$  local segment fluctuation time;

 $\tau_R$  longest Rouse relaxation time)

T. N. Khazanovich, Polymer Sci. USSR 4 (1963) 727 N. Fatkullin, R. Ki., H. W. Weber, Phys. Rev. E 47 (1993) 4600



# Renormalized Rouse formalism

## "Entangled" polymers (M>>M<sub>c</sub>)



#### many-chain problem



N. Fatkullin, R. Ki., J. Chem. Phys. 101(1994)822)





spin-lattice relaxation dispersion of polyethylene oxide melts

(R. K., N. Fatkullin, R.-O. Seitter, K. Gille, J. Chem. Phys. 98 (1998) 2173)



high mode numbers region II:  $T_1 \propto M_{\mu}^0 v^{0.25}$ 



intersegment dipolar couplings region III:  $T_1 \propto M_w^0 v^{0.45}$  Intersegment dipolar interactions:

correlation function

of the dipole pair *k*,/

 $\mathcal{G}_{kl}^{(m)}(t) = \left\langle \frac{Y_{2,m}^{*}(t)}{r^{3}(t)} \frac{Y_{2,m}(0)}{r^{3}(0)} \right\rangle$ 

probability that dipole *l* is still in  $V \propto r^3(0)$ around its

initial position





r<sub>s</sub>(†)

r(0)

**r(t**)

k

region III:  $T_1^{\text{inter}} \propto \nu^{0.4 \cdots 0.5}$ 

X

protons: intra- and intersegment interactions deuterons: only intrasegment interactions





All three T<sub>1</sub> dispersion regions are shifted into the experimental window one by one upon temperature variation

R. Ki., N. Fatkullin, R.-O. Seitter, K. Gille, J. Chem. Phys. 98 (1998) 2173

spin-lattice relaxation dispersion of polymer melts with  $M_{\mu} \square M_{c}$ ("entangled" polymers) regions I, II, III from high to low frequencies or low to high temperatures

 region I:
  $T_1 \propto M_w^{0\pm0.05} v^{0.5\pm0.05}$  

 region II:
  $T_1 \propto M_w^{0\pm0.05} v^{0.25\pm0.1}$  

 region III:
  $T_1 \propto M_w^{0\pm0.05} v^{0.45\pm0.05}$ 

a) limits b) power laws c) universal

based on data for PE, PEO, PDMS, PDES, PIB, PTHF, PS, PB regions I, II, III do <u>not</u> coincide with the Doi/Edwards limits of the reptation concept! Rouse Ren. Rouse Tube/reptation → short chains in bulk melts
→ long chains in bulk melts
→ ?

Polymer chain modes under nanoscopic constraints



1 μm

## TEM, replica

## pore width 10 nm

Linear polyethyleneoxide (PEO; M<sub>w</sub>=6000) in solid cross-linked polyhydroxyethylmethacrylate (PHEMA)

E. Fischer et al., Macromolecules 37 (2004) 3277



#### Doi/Edwards tube/reptation concept

- Curvilinear displacements along tube: s = s(t) for  $\tau_e < t < \tau_d$
- Characteristic laws for  $\langle R^2(t) \rangle$  and  $T_1(\omega)$
- Originally: Fictitious tube is to represent "chain entanglements"
- Here: Real "tube" formed by nanopores in a solid polymer matrix
- Equivalent theory, same parameters

## Predictions of the tube/reptation concept

	limits	mean squared segment displacement	spin-lattice relaxation time
(I) <sub>DE</sub>	$ au_{s} \Box (t, 1/\omega) \Box \tau_{e}$	$\langle R^2  angle \propto M^0 t^{1/2}$	$T_1 \propto -M^0 / \ln(\omega \tau_s)$
(II) <sub>DE</sub>	$\tau_{e} \Box (t, 1/\omega) \Box \tau_{R}$	$\langle R^2  angle \propto M^0 t^{1/4}$	$T_1 \propto M^0 \omega^{3/4}$
(III) <sub>DE</sub>	$ au_{R} \Box (t, 1/\omega) \Box  au_{d}$	$\langle R^2  angle \propto M^{-1/2} t^{1/2}$	$T_1 \propto M^{-1/2} \omega^{1/2}$
(IV) <sub>DE</sub>	τ <sub>d</sub> 🛛 ( <b>†,1/</b> ω)	$\langle R^2  angle \propto M^{-2} t^1$	$T_1 \propto M^{-lpha} \omega^0$

### NMR diffusometry and the tube/reptation concept

$$E\left(\vec{k},t\right) = \left\langle e^{i\vec{k}\cdot\vec{r}(t)}\right\rangle_{r} = \left\langle e^{i\vec{k}\cdot\vec{r}_{s}(t)}\right\rangle_{r_{s}} \left\langle e^{i\vec{k}\cdot\vec{r}_{c}(t)}\right\rangle_{r_{c}} \exp\left(-k^{2}D_{c}t\right)$$

$$E_{s}\left(\vec{k},t\right) = \left\langle e^{i\vec{k}\cdot\vec{r}_{s}(t)}\right\rangle_{r_{s}} \exp\left(-k^{2}D_{c}t\right)$$

$$(wave vector \ \vec{k} = \gamma \vec{G}\tau\right)$$

$$(wave vector \ \vec{k} = \gamma \vec{G}\tau)$$

$$E_{s}\left(\vec{k},t\right) = \left\langle \int \left(\frac{4\pi}{3}d_{t}|s|\right)^{-3/2}e^{r_{s}^{2}/d_{t}}e^{i\vec{k}\cdot\vec{r}_{s}(t)}d^{3}\vec{r}_{s}\right\rangle_{r_{s}} = \exp\left\{\frac{k^{4}d_{t}^{2}\left\langle s^{2}(t)\right\rangle}{72}\right\} \operatorname{erfc}\left\{\frac{k^{2}d_{t}\sqrt{\left\langle s^{2}(t)\right\rangle}}{6\sqrt{2}}\right\}$$

$$\operatorname{average over all } s$$

• mean square curvilinear segment displacements

$$\left\langle s^{2}(t) \right\rangle = \frac{2D_{0}t}{N + \frac{12d_{t}^{2}D_{0}t}{N^{2}b^{4}}} + \frac{2b\sqrt{D_{0}t}}{\sqrt{3\pi} + 18\frac{\sqrt{D_{0}t}}{Nb}}$$

 $d_{t}$  tube diameter (b, N, D<sub>0</sub> known) typical echo attenuation curves measured in linear PEO ( $M_w$ =11,200) confined in PHEMA pores at 80°C (fringe field technique; 60 T/m; 200 MHz)



echo attenuation formalism: (N. Fatkullin, R. Ki., Phys. Rev. E 52 (1<u>995) 3273)</u>

1 fitting parameter: pore diameter  $d_{pore} = (8+/-1)$  nm solid matrix consists of cross-linked hydroxyethyl-methacrylate (= HEMA)



and dodecyl-methacrylate (= DMA)



finite pore width |d<sub>p</sub> → 10 nm !)

#### Field-cycling NMR relaxometry at 85°C





R. Ki., R. O. Seitter, U. Beginn, M. Möller, N. Fatkullin, Chem. Phys. Letters 307 (1999) 147



A. Denissov, M. Kroutieva, N. Fatkullin, R. Ki., J. Chem. Phys. 116 (2002) 5217

### experimental findings



the "corset effect"

## The system dimensions

mobile linear polyethylene oxide:

PEO 2,000:  $R_F$  = 4 nm PEO 10,000:  $R_F$  = 9 nm

nearest neighbor distance 0.5 nm

rigid crosslinked HEMA+DMA methacrylate matrix:

pore diameters from 8 to 60 nm

that is: ... up to 122 PEO diameters ... up to 15 PEO radii of gyration



## the corset effect - a finite size phenomenon

conformational changes require fluctuations of the free volume
 fluctuations of the number of segments in the available volume

$$< n^2 > - < n >^2 = k_B T \rho_m \kappa_T < n >$$

#### compressibility

<n> small > segments can only be displaced along the contour line of the chain

effective tube diameter

$$d_{tube} \approx \sqrt{b^2 \rho_m k_B T \kappa_T}$$

bulk dynamics for  

$$d_{pore} \Box \left( \frac{b^3}{k_B T \kappa_T} \right)^{1/3} R_F \approx 10 R_F$$

### <u>Summary</u>

## "bulk dynamics" versus (nanoscopically) "confined dynamics"

polymer melts, *M<sub>w</sub><M<sub>c</sub>*, bulk, Rouse dynamics

polymer melts,  $M_w > M_c$ , bulk, Renormalized Rouse dynamics

rigid wall is sensed over at least 60 polymer chain diameters "corset effect":

impenetrable pore walls

- + excluded volume of polymer chains
- + low compressibility

polymer melts, *M*<sub>w</sub> arbitrary, nanoscopically confined, reptation dynamics

"tube": = "pore" for 10<sup>-2</sup> ... 3\*10<sup>-1</sup> s (8 ... 60 nm) = "tight tube" for 10<sup>-9</sup> ... 10<sup>-4</sup> s (~0.6 nm)

predicted confined-to-bulk crossover for pore diameters >> 10 R<sub>F</sub>

#### **Collaborators**:

Esteban Anoardo Ioan Ardelean **Uwe Beginn** Bogdan Buhai Alexei Denissov German Farrher Nail Fatkullin **Elmar Fischer Ravinath Kausik** Elke Kossel Margarita Kroutieva **Carlos Mattea** Hernan Tiraboschi



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Deutsche Forschungsgemeinschaft Alexander-von-Humboldt Foundation Volkswagen Foundation Recent review articles from our group:

#### a) <u>Diffusion Methods</u>

- I. Ardelean, R. Kimmich, Ann. Rep. NMR Spectr. 49 (2003) 43
- I. Ardelean, R. Kimmich, Isr. J. Chem. 43 (2003) 9

#### b) Field-cycling NMR Relaxometry

R. Kimmich, E. Anoardo, Progr. NMR Spectroscopy, 2004, in press

#### c) <u>Polymer Dynamics</u>

R. Kimmich, N. Fatkullin, Advan. Polym. Sci. 170 (2004)

#### d) <u>General Dynamics</u>

R. Kimmich, Chem. Phys. 284 (2002) 253





## polymer chain dynamics:

## polymer melts in bulk





polymer melts confined in nanoscopic pores in a solid, impenetrable matrix

#### Complex systems $\rightarrow$ complex motions $\rightarrow$ relaxation?

#### <u>examples:</u>

liquid crystals

 $\rightarrow$  restr. rot. diff. of individual molecules + collective modes

- porous media
   → restr. rot. diff. of molecules + surface diffusion (RMTD)
- polymers

 $\rightarrow$  local segment fluctuations + chain modes



## different time scale limit:

if motional components occur on very different time scales (implies stochastic independence!), then

# **Tube/Reptation concept**

Plots of the familiar BPP relaxation formulas for a single process with an exponential correlation function:



subdiffusive anomalous diffusion:

$$\langle r^2 
angle \propto t^{\kappa} \qquad (\kappa < 1)$$

- a) "(mutual) obstruction effect";
   Gaussian propagator, D=D(t)
   (e.g. single-file diffusion in zeolites, Rouse mode based diffusion)
- a) "trapping effect"; non-Gaussian propagator; waiting time distribution due to "traps" (e.g. random walk on fractals, reptation)



reptation:

- $\rightarrow$  "trapping effect"
- $\rightarrow$  non-Gaussian propagators
- → special evaluation theory for spin echo attenuation required!
- → Elmar Fischer



## deuteron field-cycling relaxometry

de Gennes' prediction of 1971 for the tube/reptation concept:

 $T_1 \propto M^0 \omega^{3/4}$ 

experiment:

 $T_1 \propto M_{PEO}^{0\pm0.05} \omega^{0.75\pm0.02}$ 

R. Ki., R. O. Seitter, U. Beginn, M. Möller, N. Fatkullin, Chem. Phys. Letters 307 (1999) 147