

## MÖSSBAUER EFFECT STUDY OF ATOMIC DISORDER IN Fe<sub>2</sub>TiSn ALLOY

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**Abstract.** The Heusler-type alloy Fe<sub>2</sub>TiSn was investigated in the temperature range (80-300) K by <sup>57</sup>Fe and <sup>119</sup>Sn Mössbauer spectroscopy. The purpose of this study was to gain information on the local atomic disorder in this alloy with an emphasis on the hyperfine parameters: magnetic field ( $B_{hf}$ ) and isomer shift (IS). The first-principle electronic structure calculations based on the tight-binding linearized muffin tin orbital (TB-LMTO) method were performed to get values of hyperfine parameters for ordered and locally disordered Fe<sub>2</sub>TiSn alloy. The analysis of <sup>57</sup>Fe Mössbauer spectra revealed the existence of an isostructural phase transition at ~240 K, while the analysis of <sup>119</sup>Sn Mössbauer spectra provides additional evidence of a change in the atomic magnetic structure near the temperature 200 K.

### 1. INTRODUCTION

The influence of the local chemical disorder on the magnetic properties of ordered Heusler-type alloys is an intriguing phenomenon. Recently in the series of paper [1-5] was shown that the Heusler Fe<sub>2</sub>TiSn alloy belongs to the group of system in which the weakly ferromagnetic and probably heavy fermion behavior is induced by atomic disorder. The Fe<sub>2</sub>TiSn crystallize in the L2<sub>1</sub>-type structure, which consists of four interpenetrating fcc sublattices. Every Fe atom has local tetrahedral symmetry and is surrounded by 4 Ti and 4 Sn atoms. Ti and Sn atoms possess octahedral local symmetry and are surrounded by 8 Fe atoms. The electronic structure LMTO calculations [1, 4] indicated that ordered Fe<sub>2</sub>TiSn should be a nonmagnetic semimetal with a pseudogap in the density of states (DOS) at the Fermi level.

The calculations predicted that the stoichiometric compound at equilibrium volume has no magnetic solution, whereas the disordered alloy in which Fe and Ti are intermixed should be magnetically ordered. The electrical resistivity, magnetization and specific heat measurements on the quenched from temperature 1023 K Heusler Fe<sub>2</sub>TiSn alloy revealed the occurrence of weak ferromagnetism [1]. At 240 K an abrupt drop in lattice parameter ( $a$ ) dependence *vs.* temperature was observed and interpreted as evidence of an isostructural phase transition, which is created by an atomic disorder. This drop coincides with unusual features in  $\chi(T)$  curve [1]. The inspection of the  $\chi(T)$  data from 250 K to 350 K leads to a Curie-Weiss law  $\chi(T) = C/(T - \theta) + \chi_0$ , where  $\theta = 250$  K. The X-ray diffraction data of Fe<sub>2</sub>TiSn revealed the presence of crystallographic disorder, which is realized by the occupation of the Ti site by Fe in every fifth unit cell [1]. The interaction between such types of "magnetic defects" *via* the long-range Ruderman-Kittel-Kasuya-Yosida mechanism may explain the observed weak ferromagnetism.

In the present paper <sup>57</sup>Fe and <sup>119</sup>Sn Mössbauer spectroscopy was used to investigate an effect of local atomic disorder in Fe<sub>2</sub>TiSn. The hyperfine parameters  $B_{hf}$  and IS were determined as a function of temperature in the range of (80-300) K. The electronic structure was calculated by LMTO method to gain better understanding of experimental results.

## 2. EXPERIMENTAL AND CALCULATION DETAILS

A Fe<sub>2</sub>TiSn ingot was prepared by arc melting the constituent metals on the water-cooled copper hearth in a high-purity argon atmosphere. The sample was annealed at 1073 K for 10 hours and then quenched in water. The <sup>57</sup>Fe and <sup>119</sup>Sn resonance absorption measurements were carried out using a constant-acceleration spectrometer. A <sup>57</sup>CoRh (50 mCi) and a Ba<sup>119m</sup>SnO<sub>3</sub> (5 mCi) sources kept at room temperature were used. A palladium foil of 0.25 mm thickness was used as a critical absorber for tin X rays. In all cases, polycrystalline absorbers with thickness ~18 mg/cm<sup>2</sup> were used. The iron and tin Mössbauer spectra were recorded from 80 to 300 K in a liquid nitrogen cryostat. The stability of the sample temperature was about 0.5 K. The special attention was done to the data analysis. The Mössbauer spectra were fitted with a least-squares method program using the Gauss-Legendre technique for evaluation of the transmission integral.

The electronic structure of the ordered compounds was studied by the all-electron self-consistent Linearized Muffin-Tin Orbital (LMTO) method and the calculations were performed using the TB~LMTO-4.7 code [6]. To test the reliability of the approximate TB~LMTO results and to investigate the effect of the Coulomb correlation interaction within the Fe-3*d* band states the electronic structure of the stoichiometric Fe<sub>2</sub>TiSn was studied by the general potential (full potential) Linear Augmented Plane Wave (FP-LAPW) method. The calculations were performed using the WIEN2K code [7].

The electronic structure of the Fe<sub>2</sub>TiSn with off-stoichiometric composition and anti-site atom defects was calculated on the supercell with translation vectors (110); ↓ including eight formula units – composition Fe<sub>16</sub>Ti<sub>8</sub>Sn<sub>8</sub>. Two kinds of defects were analyzed. In the off-stoichiometric composition the excess Fe atom was considered to occupy the nominally Ti position – the composition Fe<sub>17</sub>Ti<sub>7</sub>Sn<sub>8</sub>. The another considered defect consists of a single pair of Fe and Ti atoms at anti-site positions (Fe<sub>AS</sub> ↔ Ti<sub>AS</sub>) – composition [Fe<sub>15</sub>Ti<sub>AS</sub>][Ti<sub>7</sub>Fe<sub>AS</sub>]Sn<sub>8</sub>. In both cases one defect occur per eight unit cells.

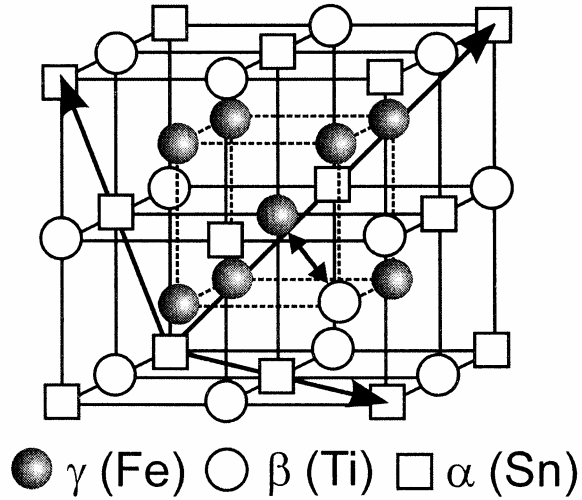
The isomer shift was calculated with the use of formula;  $IS = \alpha \Delta\rho(0)$ , where  $\alpha$  is a constant dependent on Mössbauer nuclide type and  $\Delta\rho(0)$  is a relative electron density on the nucleus area. In this paper the values:  $\alpha = -0.24 a_0^{-3}$  for <sup>57</sup>Fe nucleus [8] and  $\alpha = 0.065 a_0^{-3}$  for <sup>119</sup>Sn nucleus [9] were used. A relativistic change  $\Delta\rho(0)$  for different compositions has been calculated in relation to the value  $\rho(0)$  for  $\alpha$ -Fe and metallic Sn at 0 kbar. The isomer shift has been calculated utilizing the electronic charge density  $\rho(0)$  taken from the nucleus surface [10]. The hyperfine field  $B_{hf}$  affecting the nucleus is superposition of different contributions, but the predominant contribution arises from the Fermi contact interactions. The results for the Fermi contact hyperfine field presented in this paper were calculated by means of the relativistic formula derived by Blügel et al. [11].

## 3. RESULTS AND DISSCUSION

### A. Electronic structure calculations

In the supercell used in the electronic structure calculations, composed with the 8 unit cells of L2<sub>1</sub>-type, the Fe ↔ Ti intermixing causes, that Fe atom (denoted as Fe0) has the trigonal symmetry and is surrounded by 7 Fe atoms and 1 Ti atom in the first coordination sphere and 6 Sn atoms in the second coordination sphere (Fig. 1). The seven another Fe atoms (denoted as Fe1) have neighborhood (1Fe + 3Ti + 4Sn) in the first coordination sphere. The eighth of the last

Fig. 1. Atomic environment of the Fe sites in the eightfold supercell with L2<sub>1</sub> symmetry. The (↔) arrow shows the direction of Fe-Ti intermixing



iron atoms (denoted as Fe2) have the neighborhood (4Ti + 4Sn). The interesting result which comes from calculations is that redistribution of Fe and Ti atoms in the second coordination sphere of an Fe atom causes very little changes in the values of  $B_{hf}$  and IS. The magnetic moment and averaged values of calculated hyperfine parameters for  $^{57}\text{Fe}$  nucleus are presented in Table I.

Table I. Magnetic moment  $\mu$  and averaged values of  $B_{hf}$  and IS hyperfine parameters for  $^{57}\text{Fe}$  nucleus calculated for Fe<sub>16</sub>Ti<sub>8</sub>Sn<sub>8</sub> alloy with a single pair of Fe and Ti atoms at anti-site positions (Fe<sub>AS</sub> ↔ Ti<sub>AS</sub>)

Atom Fe	Magnetic moment [ $\mu_B$ ]	Isomer Shift [mm/s]	Hyperfine field [kGs]	Population in the supercell
Fe0	2.67	0.18	201.8	1
Fe1	0.3	0.36	40	7
Fe2	0.02	0.38	5	8

The value of IS = 0.41 mm/s (relative to  $\alpha\text{Fe}$ ) for  $^{57}\text{Fe}$  nucleus in ordered nonmagnetic Fe<sub>2</sub>TiSn compound was estimated. The local configuration of the Sn atoms located in the supercell is affected also by Fe ↔ Ti intermixing. The one Ti atom situated in the first coordination sphere of an Sn atom in the ordered L2<sub>1</sub> structure causes that a magnetic field with value  $B_{hf} \approx 5$  kGs is acted on  $^{119}\text{Sn}$  nucleus. However a single Fe atom situated in the second coordination sphere of an Sn atom in L2<sub>1</sub> structure increases the value of  $B_{hf}$  to  $\approx 55$  kGs. The value of IS = 1.6 mm/s (relative to BaSnO<sub>3</sub>) for  $^{119}\text{Sn}$  nucleus in ordered nonmagnetic Fe<sub>2</sub>TiSn compound and value of IS  $\approx 1.7$  mm/s for configurations with Fe ↔ Ti intermixing was estimated.

### B. $^{57}\text{Fe}$ and $^{119}\text{Sn}$ Mössbauer Study

The  $^{57}\text{Fe}$  Mössbauer spectrum of Fe<sub>2</sub>TiSn recorded at room temperature is shown in Fig. 2a. A single broad and asymmetric absorption line is observed. When an external magnetic field produced by a rare earth permanent magnet with intensity of 0.42 T is applied, the additional broadening of absorption line is observed, suggesting the presence of a magnetic component

(Fig. 2b). Taking into account results of electronic structure we performed fit to the spectra assuming existence of a magnetic component (Zeeman sextet) that comes from configuration of Fe atoms situated on Ti sites, and a non-magnetic (single line) component arising from ordered bulk.

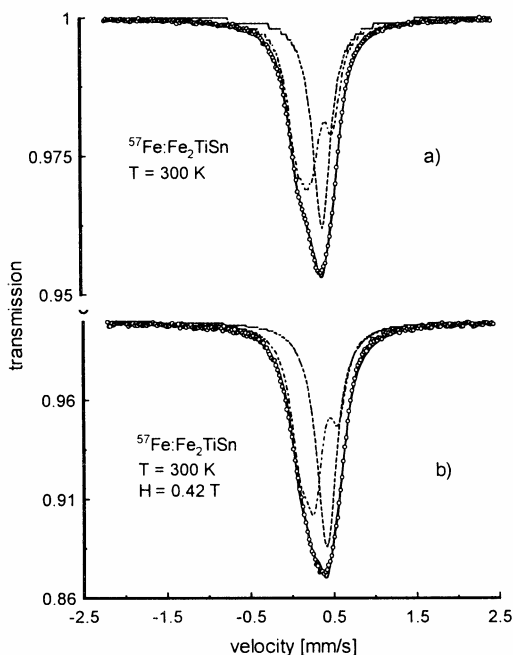


Fig. 2. Mössbauer spectra of  $^{57}\text{Fe} : \text{Fe}_2\text{TiSn}$  measured at room temperature: (a) in zero-applied field, (b) in an external magnetic field of 0.42 T perpendicular to the direction of  $\gamma$ -radiation

For sample measured in zero-applied field and room temperature the following parameters of Zeeman sextet was estimated: hyperfine magnetic field  $\text{HFM} = 10.2$  kGs, quadrupole splitting  $\text{QS} = 0.14$  mm/s and isomer shift  $\text{IS} = 0.21$  mm/s. The isomer shift for single line was  $\text{IS} = 0.35$  mm/s. The measurements performed at 80 K and in velocity range appropriate for detecting iron with high magnetic moment (F0 atom) reveal that there are no absorption lines located at  $|\nu| > 1$  mm/s. Although intensity of the high magnetic field component should be small (see Table I), the observed absence of such component in measured spectra indicates that distance between neighboring Fe0 atoms is longer than five lattice constants, in opposition to the results presented in [1].

In the Fig. 4 the temperature dependence of isomer shift for magnetic ( $\text{IS}_M$ ) and non-magnetic ( $\text{IS}_{NM}$ ) components is presented. The two features are interesting. The observed at 240 K a small drop in  $\text{IS}_{NM}$  curve coincides with observed abrupt drop in the experimental  $\alpha(T)$  plot [1]. Although the drop in  $\text{IS}_{NM}(T)$  is within the errors estimated from the fitting procedure can be interpreted as an evidence of a structural transition in the paramagnetic phase. The transition is probably involved by presence of the monoclinic or orthorhombic deformation of the  $\text{L2}_1$  structure of the investigated  $\text{Fe}_2\text{TiSn}$  samples. The comparison of the calculated charge density distributions within the unit cells of  $\text{L2}_1$  and orthorhombic structures have shown that the orthorhombic distortion increases the values of charge densities at nuclei of each constituent atom:  $+0.39$  (a.u.) $^{-3}$  for Sn,  $+0.17$  (a.u.) $^{-3}$  for Fe and  $+0.06$  (a.u.) $^{-3}$  for Ti. It gives the net change in

Fig. 3. Mössbauer spectra of  $^{119}Sn: Fe_2TiSn$  measured at room temperature: (a) in zero-applied field, (b) in an external magnetic field of 0.42 T perpendicular to the direction of  $\gamma$ -radiation

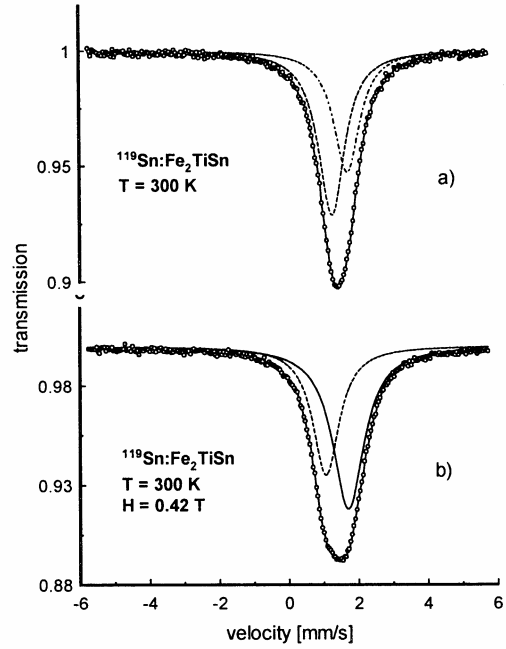
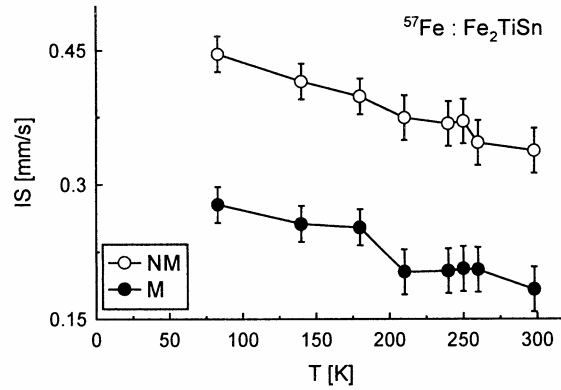


Fig. 4. The temperature dependence of the  $^{57}Fe$  isomer shift for magnetic (M) and non-magnetic (NM) components of  $Fe_2TiSn$  Mössbauer spectra. Lines connecting experimental points are added as a guide to the eye



$^{57}Fe$ - $IS_{NM}$  about 0.04 mm/s, with the good relation to experiment. The drop in  $IS_M$  curve is shifted to the lower temperatures and is observed in the region near 200 K. It is interesting that this drop coincides with same features in  $B_{hf}(T)$  curve, presented on Fig. 5. An increase in  $B_{hf}$  at the region of (200-240) K is observed. The physical nature of this behavior is under investigations. Probably a micromagnetic feature of the specific type of “magnetic defects” which are distributed in disordered  $Fe_2TiAl$  alloys is established in this interesting region.

The  $^{119}Sn$  Mössbauer spectrum of  $Fe_2TiSn$  recorded at room temperature is shown in Fig. 3a. A single absorption line is observed, which broadening increases in an external field with intensity of 0.42 T (Fig. 3b). Taking into account the features of  $^{119}Sn$  Mössbauer spectroscopy (the thick natural width of the resonant line), it is difficult to perform fitting of the single line spectra with assumption of existence of a magnetic component. We performed deconvolution of

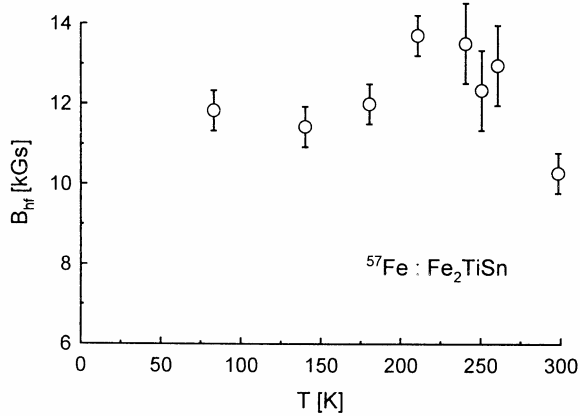


Fig. 5. The temperature dependence of the  $^{57}\text{Fe}$  hyperfine magnetic field  $B_{hf}$  for magnetic component

$^{119}\text{Sn}$  Mössbauer spectra on the two Lorentzian components in the investigated region of temperature (80-300) K. The first one has a stable full widths at half maximum FWHM = 0.95 mm/s. The second Lorentzian line has the IS value in the range (1.66-1.71) mm/s and growing FWHM from 0.92 mm/s at 300 K to 1.34 mm/s at 80 K. The first component can be attributed to the ordered non-magnetic  $\text{Fe}_2\text{TiSn}$  bulk, while the second component to the magnetic clusters produced by Fe - Ti intermixing. The value of 9.2 kGs for  $B_{hf}$  acting at  $^{119}\text{Sn}$  nucleus at 80 K was estimated from a transmission integral fitting procedure. The values: IS = 1.66 mm/s for magnetic component and IS = 1.52 mm/s for non-magnetic components were determined, with the good accordance to the theoretical ones. The dependence of the area under resonant lines *versus* temperature which correspond to temperature dependence of the Mössbauer absorption probability  $f'$  is presented on Fig. 6. A non-linear variation of the  $^{119}\text{Sn}$  absorption probability is observed with

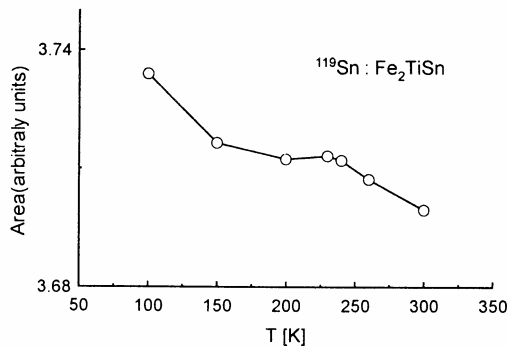


Fig. 6. The area under the  $^{119}\text{Sn}$  resonant spectrum of  $\text{Fe}_2\text{TiSn}$  alloy plotted *versus* the temperature. Lines connecting experimental points are added as a guide to the eye

a significant drop at 200 K. It provides additional evidence of a change in the atomic magnetic structure near the temperature 200 K.

#### 4. CONCLUSIONS

The presented results indicate that the Mössbauer spectroscopy is a powerful tool to characterize influence of disorder on the magnetic properties of the  $\text{Fe}_2\text{TiSn}$  alloy. The performed electron

structure calculation and observed at 240 K drop in IS(T) curve for <sup>57</sup>Fe nucleus reveals evidence of a structural phase transition in the paramagnetic phase. The analysis of <sup>57</sup>Fe and <sup>119</sup>Sn Mössbauer spectra provides additional evidence of a change in the atomic magnetic structure near the temperature 200 K. This behavior is probably connected with mictomagnetism of the specific type of “magnetic defects” which are created in disordered Fe<sub>2</sub>TiAl alloys due to Fe – Ti intermixing. The values of hyperfine parameters ( $B_{hf}$  and IS) obtained from electronic structure calculations are in the good relation to the experimental data.

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